



(12)

EUROPEAN PATENT APPLICATION

(43) Date of publication:

20.09.2000 Bulletin 2000/38

(51) Int. Cl.:

H01L 21/768, H01L 21/762

(21) Application number: 00103333.1

(22) Date of filing: 21.02.2000

(84) Designated Contracting States:

AT BE CH CY DE DK ES FI FR GB GR IE IT LI LU
MC NL PT SEDesignated Extension States:
AL LT LV MK RO SI

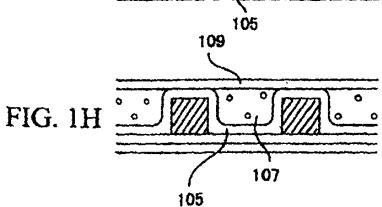
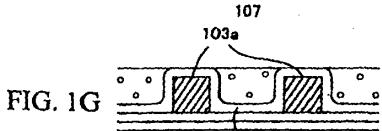
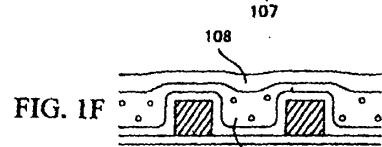
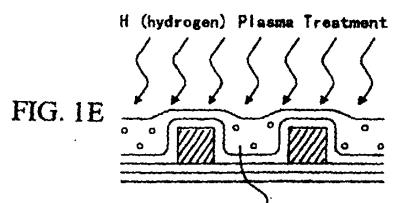
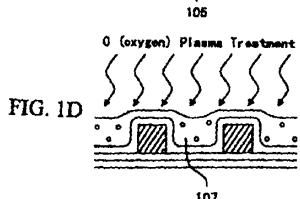
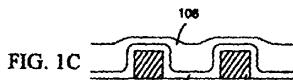
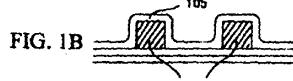
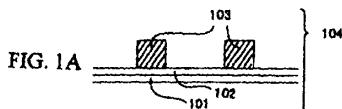
(30) Priority: 17.03.1999 JP 7225299

(71) Applicants:

, Canon Sales Co., Inc.
Minato-ku, Tokyo 108-0073 (JP), Semiconductor Process Laboratory Co., Ltd.
Minato-ku, Tokyo 108-0075 (JP)(72) Inventor: Maeda, Kazuo, Semiconductor Process
Lab. Co., Ltd.
Tokyo 108-0075 (JP)(74) Representative:
Schwabe - Sandmair - Marx
Stuntzstrasse 16
81677 München (DE)

(54) Method for forming a porous silicon oxide film

(57) A method for forming an interlayer insulating film is disclosed. This method comprises the steps of: forming an Si-C film or an Si-C-H film on an underlying insulating film by performing plasma polymerization for an Si and C containing compound; forming a porous SiO_2 film by performing O (oxygen) plasma oxidation for the Si-C film or the Si-C-H film; and forming a cover insulating film on the porous SiO_2 film by performing H (hydrogen) plasma treatment for the porous SiO_2 film.



Description

BACKGROUND OF THE INVENTION

5 [0001] The present invention relates to a method for forming an interlayer insulating film and, more particularly, to a method for forming an interlayer insulating film having a low dielectric constant, which is necessary for a highly-integrated semiconductor device. A progress in high integration regarding the semiconductor device in recent years has resulted in a narrower interval between wiring lines. As the narrowed interval between the wiring lines causes an increase in capacitance between the wiring lines, a request has been made for formation of an interlayer insulating film, which has a low dielectric constant.

10 [0002] With recent progresses in high integration of an LSI device, the wiring line has been micronized and multilayered. There has also been an increase in capacitance between the wiring lines. Such an increase in capacitance has caused a great reduction in an operating speed. Thus, improvement in this regard has been strongly demanded. As one of improvement measures, a method for reducing capacitance between the wiring lines has been studied. This method uses an interlayer insulating film, which has a dielectric constant lower than that of SiO_2 currently used for an interlayer insulating film.

15 [0003] Typical interlayer insulating films of low dielectric constants currently under study are ① an SiOF film, and ② an organic insulating film of a low dielectric constant. Description will now be made of these films.

① SiOF Film

20 [0004] An SiOF film is formed by using source gas containing F and substituting Si-F bond for a portion of Si-O bond in SiO_2 . This SiOF film has a relative dielectric constant, which is monotonically reduced as concentration of F in the film increases.

25 [0005] For forming such SiOF films, several methods have been reported (see p.82 of monthly periodical "Semiconductor World", February issue of 1996). Most promising among these methods is one for forming an SiOF film by using SiH_4 , O_2 , Ar and SiF_4 as source gases, and by a high-density plasma enhanced CVD method (HDPCVD method). A relative dielectric constant of an SiOF film formed by this method is in a range of 3.1 to 4.0 (varies depending on F concentration in the film). This value is lower than a relative dielectric constant 4.0 of SiO_2 , which has conventionally been used for the interlayer insulating film.

② Organic Insulating Film of Low Dielectric Constant

30 [0006] As an insulating film which has a lower dielectric constant (3.0 or lower) compared with the SiOF film, an organic insulating film of a low dielectric constant is now a focus of attention. Table 1 shows a few organic insulating films of low dielectric constants, which have been reported, and respective relative dielectric constants and thermal decomposition temperatures thereof.

TABLE 1

| | Organic Insulating Film | Relative Dielectric Constant | Thermal Decomposition Temperature (°C) | Note |
|----|---------------------------|------------------------------|----------------------------------------|---------------------------------------------------------------------------|
| | Fluorine-containing resin | 2.4 | 420 | p.82 of monthly periodical "Semiconductor World", February issue of 1997 |
| 40 | Cytop | 2.1 | 400 | p. 90 of monthly periodical "Semiconductor World", February issue of 1996 |
| 45 | Amorphous telon | 1.9 | 400 | p. 91 of monthly periodical "Semiconductor World", February issue of 1996 |

50 [0007] However, the SiOF film is disadvantageous in that an increase in concentration of F in the film leads to a reduction in moisture absorption resistance. The reduced moisture absorption resistance poses a serious problem, because a transistor characteristic and adhesion of an upper barrier metal layer are affected.

55 [0008] Peeling-off easily occurs in the organic insulating film of a low dielectric constant, because of bad adhesion with a silicon wafer or the SiO_2 film. Furthermore, the organic insulating film is disadvantageous in that heat resistivity is low since a thermal decomposition temperature is around 400 °C. The disadvantage of low heat resistivity poses a problem for annealing a wafer at a high temperature.

SUMMARY OF THE INVENTION

[0009] It is an object of the present invention to provide a method for forming an interlayer insulating film of a low dielectric constant, which has good moisture absorption resistance and heat resistivity. It is another object of the invention to provide a semiconductor device, which employs the above method.

[0010] Description will now be made of an interlayer insulating film of the present invention by referring to Table 2.

5

10

15

20

25

| Porous Film | Source Gas | Film by Plasma Polymerization |
|------------------------------------|---------------------------------------------------------|----------------------------------|
| SiO ₂ film | TEOS | Si-C film Si-C-H film |
| | TEOS + O ₂ | Si-C-O film Si-C-O-H film |
| | TEOS + H ₂ O | Si-C-O-H film |
| B-containing SiO ₂ film | TEOS + B ₂ H ₆ | Si-C-B film Si-C-B-H film |
| | TEOS + B ₂ H ₆ + O ₂ | Si-C-B-O film Si-C-B-O-H film |
| | TEOS + B ₂ H ₆ + H ₂ O | Si-C-B-O-H film |
| F-containing SiO ₂ film | TEOS + C ₂ F ₆ | Si-C-F film Si-C-F-H film |
| | TEOS + C ₂ F ₆ + O ₂ | Si-C-F-O film Si-C-F-O-H film |
| | TEOS + C ₂ F ₆ + H ₂ O | Si-C-F-O-H film |

[0011] For formation of a porous SiO₂ film of the present invention, TEOS, TEOS + O₂ or TEOS + H₂O is used as source gas. By performing plasma polymerization for such source gas, an Si-C film, an Si-C-H film, an Si-C-O film or an Si-C-O-H film is formed on a formed body. Then, by performing O (oxygen) plasma treatment for these films, C or H is oxidized in the film. C or H is oxidized in the film, and voids are formed in portions from which C or H has been discharged. Accordingly, a porous SiO₂ film is formed. A porous SiO₂ film can also be formed by using methylsilane (Si(CH₃)H₃), instead of TEOS.

[0012] For formation of a porous B(boron)-containing SiO₂ film of the present invention, TEOS + B₂H₆, TEOS + B₂H₆ + O₂ or TEOS + B₂H₆ + H₂O is used as source gas. By performing plasma polymerization for such source gas, an Si-C-B film, an Si-C-B-H film, an Si-C-B-O film or an Si-C-B-O-H film is formed on a formed body. Then, by performing O (oxygen) plasma treatment for these films, C or H is oxidized in the film. C or H is oxidized in the film, and voids are formed in portions from which C or H has been discharged. Accordingly, a porous B-containing SiO₂ film is formed. A porous B-containing SiO₂ film can also be formed by using methylsilane (Si(CH₃)H₃) or trimethylsilaborane ((CH₃)₃SiO₃B), instead of TEOS in source gas. Instead of B₂H₆ in source gas, TMB(B(OCH₃)) or TEB(B(OC₂H₅)) can be used to form a porous B-containing SiO₂ film.

[0013] For formation of a porous F-containing SiO₂ film of the present invention, TEOS + C₂F₆, TEOS + C₂F₆ + O₂ or TEOS + C₂F₆ + H₂O is used as source gas. By performing plasma polymerization for such source gas, an Si-C-F film, an Si-C-F-H film, an Si-C-F-O film or an Si-C-F-O-H film is formed on an object to be formed. Then, by performing O (oxygen) plasma treatment for these films, C or H is oxidized in the film. C or H is oxidized in the film, and voids are formed in portions from which C or H has been discharged. Accordingly, a porous F-containing SiO₂ film is formed. A porous F-containing SiO₂ film can also be formed by using methylsilane (Si(CH₃)H₃), instead of TEOS.

BRIEF DESCRIPTION OF THE DRAWINGS

50

[0014]

FIGS. 1A to 1H are cross - sectional views respectively showing methods for forming interlayer insulating films according to 1st, 3rd, 5th, 7th, 9th, 11th and 13th embodiments of the present invention; and

55 FIGS. 2A to 2M are cross-sectional view respectively showing methods for forming interlayer insulating films according to 2nd, 4th, 6th, 8th, 10th, 12th and 14th embodiments of the invention.

DESCRIPTION OF THE PREFERRED EMBODIMENTS

[0015] Next, description will be made of the preferred embodiments of the present invention with reference to the accompanying drawings.

5 (First Embodiment)

[0016] FIGS. 1A to 1H are cross-sectional views illustrating a first embodiment of the present invention.

[0017] First, as shown in FIG. 1A, a BPSG (borophosphosilicate glass) film 102 is formed on a silicon substrate 101. Then, after an aluminum film is formed on the BPSG film 102, an aluminum wiring layer 103 is formed by patterning the aluminum film. The silicon substrate 101, the BPSG film 102 and the aluminum wiring layer 103 formed in this manner constitute an object 104 to be formed.

[0018] Then, as shown in FIG. 1B, an SiO₂ film 105 (underlying insulating film) is formed on the object 104 to be formed. This SiO₂ film 105 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases. A film thickness of this SiO₂ film 105 is 100 nm. This SiO₂ film 105 can prevent H₂O from being dispersed in the aluminum wiring layer 103, which may cause corrosion.

[0019] Subsequently, as shown in FIG. 1C, a carbon or carbon and hydrogen containing (referred to as Si-C (-H), hereinafter) film 106 is formed on the SiO₂ film 105 (underlying insulating film). This Si-C (-H) film 106 is formed by using 50 sccm TEOS (Tetra-Ethyl-Ortho-Silicate) as source gas, applying an RF power having a frequency of 13.56 MHz and a power of 100 W while keeping the silicon substrate 101 at 100 °C, and performing plasma polymerization for TEOS at pressure of 1 Torr. A film thickness of this Si-C (-H) film 106 is 500 nm. It should be noted that, in forming the Si-C (-H) film 106, one of methylsilane (Si(CH₃)₃) and trimethylsilylborate ((CH₃)₃SiO₃B) can also be used instead of the TEOS. When using one of these gases, the flow rate of the gas is 50 sccm and other process conditions are the same as in the case using the TEOS. Using the trimethylsilylborate, the Si-C (-H) film 106 further contains B (boron).

[0020] Then, as shown in FIG. 1D, O (oxygen) plasma treatment is performed for the Si-C (-H) film 106. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 101 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0021] In this O (oxygen) plasma treatment C or H contained in the Si-C (-H) film 106 is oxidized, and discharged to the outside of the film. Voids are formed in portions from which C or H has been discharged, and Si-O bond is formed in the portions. Thus, the Si-C (-H) film 106 becomes a porous SiO₂ film 107.

[0022] Subsequently, as shown in FIG. 1E, H (hydrogen) plasma treatment is performed for the porous SiO₂ film 107.

[0023] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 101 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

[0024] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0025] Subsequently, as shown in FIG. 1F, an SiO₂ film 108 is formed on the porous SiO₂ film 107. This SiO₂ film 108 is formed by a CVD method, which contains TEOS and O₃ as source gases. O₃ contained as source gas has concentration enough for oxidation of TEOS. Accordingly, the SiO₂ film 108 shows flowability, and can be planarized to a considerable extent.

[0026] Then, as shown in FIG. 1G, the SiO₂ film 108 is polished by a CMP method (chemical mechanical polishing method) to planarize its surface. At this time, sane portions of the previously formed SiO₂ fibre 105 and 107 are eliminated by polishing. The planarization by the CMP method should be carried out to prevent complete elimination of the SiO₂ film 105 formed on a convexity 103a of the aluminum wiring layer.

[0027] Subsequently, as shown in FIG. 1H, an SiO₂ film 109 (cover insulating layer) is formed on the planarized surface. This SiO₂ film 109 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. A film thickness of this SiO₂ film 109 is 100 nm.

[0028] The foregoing process of forming the SiO₂ film 105 (underlying insulating film), 107 and 109 (cover insulating film) results in formation, on the object 104 to be formed, of an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the SiO₂ film 107 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 109 is formed on the porous SiO₂ film 107, incursion of water into the SiO₂ film 107 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 107 can improve the moisture absorption resistance of the film. Moreover, the SiO₂ films 105, 107 and 109 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

(Second Embodiment)

[0029] A second embodiment is a case of applying the first embodiment to a damascene process.

[0030] FIGS. 2A to 2M are cross-sectional views illustrating the second embodiment.

[0031] First, as shown in FIG. 2A, a BPSG (borophosphosilicate glass) film 202 is formed on a silicon substrate 201. After an aluminum layer is formed on the BPSG film 202 an aluminum wiring layer 203 is formed by patterning the aluminum layer. It should be noted that the aluminum wiring layer 203 in figures is not patterned for convenience. Then, the silicon substrate 201, the BPSG film 202 and the aluminum wiring layer 203 constitute an object 204 to be formed.

[0032] Then, as shown in FIG. 2B, an SiO_2 film 205 (underlying insulating film) having a film thickness of 100 nm is formed on the aluminum wiring layer 203. This SiO_2 film 205 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH_4 and N_2O are used as source gases.

[0033] Subsequently, as shown in FIG. 2C, an Si-C (-H) film 206 is formed on the SiO_2 film 205 (underlying insulating film). This Si-C (-H) film 206 is formed by using 50 sccm TEOS (Tetra-Ethyl-Ortho-Silicate) as source gas, applying an RF power having a frequency of 13.56 MHz and a power of 100 W while keeping the silicon substrate 201 at 100 °C, and performing plasma polymerization for TEOS at pressure of 1 Torr. A film thickness of the Si-C (-H) film 206 is 500 nm. It should be noted that, in fanning the Si-C (-H) film 206, one of methylsilane ($\text{Si}(\text{CH}_3)_2\text{H}_3$) and trimethylsilylborate ($\{(\text{CH}_3)_2\text{SiO}\}_3\text{B}$) can also be used instead of the TEOS. When using one of these gases, the flow rate of the gas is 50 sccm and other process conditions are the same as in the case using the TEOS. Using the trimethylsilylborate, the Si-C (-H) film 206 further contains B (boron).

[0034] Then, as shown in FIG. 2D, O (oxygen) plasma treatment is performed for the Si-C (-H) film 206. This O (oxygen) plasma treatment is performed supplying 600 sccm O_2 to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O_2 . The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 201 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0035] In this O (oxygen) plasma treatment C or H contained in the Si-C (-H) film 206 is then oxidized, and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portions. Accordingly, the Si-C (-H) film 206 becomes a porous SiO_2 film 207.

[0036] Subsequently, as shown in FIG. 2E, H (hydrogen) plasma treatment is performed for the porous SiO_2 film 207.

[0037] This H plasma treatment is performed by supplying H_2 of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 201 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

[0038] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0039] Subsequently, as shown in FIG. 2F, a damascene trench 208 is formed by patterning the SiO_2 film 207. This damascene trench 208 reaches the SiO_2 film 205, which has been formed below the SiO_2 film 207.

[0040] Then, as shown in FIG. 2G, an SiO_2 film 209 (first insulating film) is formed on the SiO_2 film 207 and on the side and bottom portions of the damascene trench 208. This SiO_2 film 209 is formed by a plasma enhanced CVD method, and SiH_4 and N_2O are used as source gases. By the SiO_2 film 209 formed in the side portion of the damascene trench 208, Cu buried later in the damascene trench 208 can be prevented from being dispersed inside the porous SiO_2 film 207.

[0041] Then, as shown in FIG. 2H, anisotropic etching is performed for the SiO_2 film 209 (first insulating film) and the SiO_2 film 205 (underlying insulating film) formed on the bottom portion of the damascene trench 208. Accordingly, the SiO_2 film 209 is eliminated except for a portion formed on the side portion of the damascene trench 208. Also, since the SiO_2 film 205 formed below the damascene trench 208 is eliminated, a contact hole that reaches the aluminum wiring layer 203 is formed below the damascene trench 208.

[0042] Subsequently, as shown in FIG. 2I, a Cu-plated film 210 is formed in the damascene trench 208 and on the SiO_2 film 207. The Cu-plated film 210 formed in the damascene trench 208 is used as a Cu wiring line.

[0043] Then, as shown in FIG. 2J, the Cu-plated film 210 formed on the SiO_2 film 207 is polished by a CMP method to be eliminated. Accordingly, the Cu-plated film remains only in the damascene trench 208.

[0044] Subsequently, as shown in FIG. 2K, a barrier metal TiN film 211 is formed above the damascene trench 208. Accordingly, Cu in the damascene trench 208 can be prevented from being dispersed in an SiO_2 film formed later above the damascene trench 208.

[0045] Then, as shown in FIG. 2L, patterning is performed for the TiN film 211 to leave a TiN film 211a formed above the damascene trench 208, and the TiN film 211 formed in the other portions is etched to be eliminated.

[0046] Subsequently, as shown in FIG. 2M, an SiO_2 film 212 (cover insulating film) is formed on the SiO_2 film 207 and the TiN film 211a. This SiO_2 film 212 is formed by a plasma enhanced CVD method, and SiH_4 and N_2O are used as

source gases.

[0047] The foregoing process results in formation, on the object 204 to be formed, of an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the SiO₂ film 207 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 212 (cover insulating film) is formed on the porous SiO₂ film 207, incursion of water into the SiO₂ film 207 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 207 can improve the moisture absorption resistance of the film. Moreover, the SiO₂ films 207 and 212 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

(Third Embodiment)

[0048] FIGS. 1A to 1H are cross-sectional views illustrating a third embodiment.

[0049] First, as shown in FIG. 1A, a BPSG (borophosphosilicate glass) film 102 is formed on a silicon substrate 101. Then, after an aluminum film is formed on the BPSG film 102, patterning is performed for the same to form an aluminum wiring layer 103. The silicon substrate 101, the BPSG film 102 and the aluminum wiring layer 103 formed in this manner constitute an object 104 to be formed.

[0050] Then, as shown in FIG. 1B, an SiO₂ film 105 (underlying insulating film) is formed on the object 104 to be formed. This SiO₂ film 105 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases. A film thickness of the SiO₂ film 105 is 100 nm. The SiO₂ film 105 can prevent H₂O from being dispersed in the aluminum wiring layer 103, which may cause corrosion.

[0051] Subsequently, as shown in FIG. 1C, a carbon or a carbon and hydrogen containing (referred to as Si-C (-H), hereinafter) film 106 is formed on the SiO₂ film 105 (underlying insulating film). This Si-C (-H) film 106 is formed by using SiH₄ and H₂O as source gases, applying an RF power having a frequency of 13.56 MHz and power of 300 W while keeping the silicon substrate 101 at 100 °C, and performing plasma polymerization for SiH₄ and H₂O at pressure of 1 Torr. At this time, flow rates of the source gases are 30 sccm and 60 sccm for SiH₄ and H₂O respectively. A film thickness of the Si-C (-H) film 106 is 500 nm.

[0052] It should be noted that in forming Si-C (-H) film 106 the methylsilane (Si(CH₃)H₃) can also be used instead of the SiH₄. When using the methylsilane (Si(CH₃)H₃) its flow rate is 30 sccm and the power of the RF power is 100 W, and the other process conditions are the same as in the case using the SiH₄.

[0053] Then, as shown in FIG. 1D, O (oxygen) plasma treatment is performed for the Si-C (-H) film 106. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 101 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0054] In this O (oxygen) plasma treatment C or H contained in the Si-C (-H) film 106 is oxidized and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C (-H) film 106 becomes a porous SiO₂ film 107.

[0055] Subsequently, as shown in FIG. 1E, H (hydrogen) plasma treatment is performed for the porous SiO₂ film 107.

[0056] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 101 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

[0057] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0058] Subsequently, as shown in FIG. 1F, an SiO₂ film 108 is formed on the porous SiO₂ film 107. This SiO₂ film 108 is formed by a CVD method, which employs source gas containing TEOS and O₃. In this case, since O₃ in source gas has concentration enough for oxidation of TEOS, the SiO₂ film 108 shows flowability, and can be planarized to a considerable extent.

[0059] Then, as shown in FIG. 1G, the SiO₂ film 108 is polished by a CMP method (chemical mechanical polishing method) so as to planarize its surface. At this time, some portions of the previously formed SiO₂ fibre 105 and the SiO₂ film 107 are eliminated by polishing. The planarizing by the CMP method should be carried out not to eliminate the whole SiO₂ film 105 formed on a convexity 103a of the aluminum wiring layer.

[0060] Subsequently, as shown in FIG. 1H, an SiO₂ film 109 (cover insulating film) is formed on the planarized surface. This SiO₂ film 109 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. A film thickness of the SiO₂ film 109 is 100 nm.

[0061] The foregoing process of forming the SiO₂ films 105 (underlying insulating film), 107 and 109 (cover insulating film) results in formation, on the object 104 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the SiO₂ film 107 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual

EP 1 037 275 A1

SiO₂ film 109 is formed on the porous SiO₂ film 107, incursion of water into the SiO₂ film 107 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 107 can improve the moisture absorption resistance of the film. Moreover, the SiO₂ films 105, 107 and 109 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

5 (Fourth Embodiment)

[0062] A fourth embodiment is a case of applying the third embodiment to a damascene process.

[0063] FIGS. 2A to 2M are cross-sectional views illustrating the fourth embodiment.

[0064] First, as shown in FIG. 2A, a BPSG (borophosphosilicate glass) film 202 is formed on a silicon substrate 201. Then, after an aluminum layer is formed thereon, patterning is performed for the same to form an aluminum wiring layer 203. It should be noted that the aluminum wiring layer 203 in figures is not patterned for convenience. The silicon substrate 201, the BPSG film 202 and the aluminum wiring layer 203 constitute an object 204 to be formed.

[0065] As shown in FIG. 2B, an SiO₂ film 205 (underlying insulating film) having a film thickness of 100 nm is formed on the aluminum wiring layer 203. This SiO₂ film 205 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases.

[0066] Subsequently, as shown in FIG. 2C, an Si-C (-H) film 206 is formed on the SiO₂ film 205 (underlying insulating film). This Si-C (-H) film 206 is formed by using SiH₄ and H₂O as source gases, applying an RF power having a frequency of 13.56 MHz and power of 300 W while keeping the silicon substrate 201 at 100 °C, and performing plasma polymerization for SiH₄ and H₂O at pressure of 1 Torr. At this time, flow rates of source gases are 30 sccm and 60 sccm for SiH₄ and H₂O respectively. A film thickness of the Si-C (-H) film 206 is 500 nm.

[0067] It should be noted that in forming Si-C (-H) film 206 the methylsilane (Si(CH₃)H₃) can also be used instead of the SiH₄. When using the methylsilane (Si(CH₃)H₃) its flow rate is 30 sccm and the power of the RF power is 100 W, and the other process conditions are the same as in the case using the SiH₄.

[0068] Then, as shown in FIG. 2D, O (oxygen) plasma treatment is performed for the Si-C (-H) film 206. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 201 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0069] In this O (oxygen) plasma treatment C or H contained in the Si-C (-H) film 206 is oxidized and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C (-H) film 206 becomes a porous SiO₂ film 207.

[0070] Subsequently, as shown in FIG. 2E, H (hydrogen) plasma treatment is performed for the porous SiO₂ film 207.

[0071] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 201 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

[0072] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0073] Subsequently, as shown in FIG. 2F, patterning is performed for the porous SiO₂ film 207 to form a damascene trench 208. This damascene trench 208 reaches the SiO₂ film 205 formed below the SiO₂ film 207.

[0074] Then, as shown in FIG. 2G, an SiO₂ film 209 (first insulating film) is formed on the SiO₂ film 207 and on the side and bottom portions of the damascene trench 208. This SiO₂ film 209 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. By the SiO₂ film 209 formed on the side portion of the damascene trench 208, Cu buried later in the damascene trench 208 can be prevented from being dispersed in the porous SiO₂ film 207.

[0075] Then, as shown in FIG. 2H, anisotropic etching is performed for the SiO₂ film 209 (first insulating film) and the SiO₂ film 205 (underlying insulating film) formed below the damascene trench 208. Accordingly, the SiO₂ film 209 is eliminated except for a portion formed on the side portion of the damascene trench 208. The SiO₂ film 205 formed below the damascene trench 208 is also eliminated. Thus, a contact hole that reaches the aluminum wiring layer 203 is formed below the damascene trench 208.

[0076] Subsequently, as shown in FIG. 2I, a Cu-plated film 210 is formed in the damascene trench 208 and on the SiO₂ film 207. The Cu-plated film 210 formed in the damascene trench 208 is used as a Cu wiring line.

[0077] Then, as shown in FIG. 2J, the Cu-plated film 210 formed on the SiO₂ film 207 is polished by a CMP method to be eliminated. Accordingly, the Cu-plated film remains only in the damascene trench 208.

[0078] Subsequently, as shown in FIG. 2K, a barrier metal TiN film 211 is formed above the damascene trench 208. Accordingly, Cu in the damascene trench 208 can be prevented from being dispersed in an SiO₂ film later formed above the same.

[0079] Then, as shown in FIG. 2L, patterning is performed to leave a TiN film 211a formed above the damascene

EP 1 037 275 A1

trench 208, and the TiN film 211 formed on the other portions is etched to be eliminated.

[0080] Subsequently, as shown in FIG. 2M, an SiO₂ film 212 (cover insulating film) is formed on the SiO₂ film 207 and the TiN film 211a. This SiO₂ film 212 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases.

[0081] The foregoing process results in formation, on the object 204 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the SiO₂ film 207 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 212 (cover insulating film) is formed on the porous SiO₂ film 207, incursion of water into the SiO₂ film 207 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 207 can improve the moisture absorption resistance of the film. Moreover, the SiO₂ films 207 and 212 have better heat resistivity compared with the organic insulating film film since these films consist mainly of Si and O.

(Fifth Embodiment)

[0082] A fifth embodiment is different from the first to fourth embodiments in that a carbon and oxygen containing, or carbon, oxygen and hydrogen containing (referred to as Si-C-O (-H), hereinafter) film is formed by performing plasma polymerization for TEOS and O₂.

[0083] FIGS. 1A to 1H are cross-sectional views illustrating the fifth embodiment.

[0084] First, as shown in FIG. 1A, a BPSG (borophosphosilicate glass) film 102 is formed on a silicon substrate 101. Then, after an aluminum film is formed on the BPSG film 102, patterning is performed for the same to form an aluminum wiring layer 103. The silicon substrate 101, the BPSG film 102 and the aluminum wiring layer 103 formed in this manner constitute an object 104 to be formed.

[0085] Then, as shown in FIG. 1B, an SiO₂ film 105 (underlying insulating film) is formed on the object 104 to be formed. This SiO₂ film 105 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases. A film thickness of this SiO₂ film 105 is 100 nm. The SiO₂ film 105 can prevent H₂O from being dispersed in the aluminum wiring layer 103, which may cause corrosion.

[0086] Subsequently, as shown in FIG. 1C, an Si-C-O (-H) film 106 is formed on the SiO₂ film 105 (underlying insulating film). This Si-C-O (-H) film 106 is formed by performing plasma polymerization for TEOS and O₂. This plasma polymerization is performed by applying an RF power having a frequency of 13.56 MHz and power of 100 W to the TEOS and O₂. Flow rates of the source gases at this time are 30 sccm for TEOS and 240 sccm for O₂, respectively. In forming the Si-C-O (-H) film 106 the temperature of the silicon substrate 101 is maintained at 400 °C and the pressure is held at 1 Torr. A film thickness of the Si-C-O (-H) 106 is 500 nm.

[0087] It should be noted that, in forming the Si-C-O (-H) 106, one of methylsilane (Si(CH₃)₃) and trimethylsilylborate ((CH₃)₃SiO)B can also be used instead of the TEOS. When using one of these gases, the flow rate of the gas is 30 sccm and other process conditions are the same as in the case using the TEOS. Using the trimethylsilylborate, the Si-C-O (-H) film 106 further contains B (boron).

[0088] Then, as shown in FIG. 1D, O (oxygen) plasma treatment is performed for the Si-C-O (-H) film 106. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 101 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0089] In this O (oxygen) plasma treatment C or H contained in the Si-C-O (-H) film 106 is oxidized and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C-O (-H) film 106 becomes a porous SiO₂ film 107.

[0090] Subsequently, as shown in FIG. 1E, H (hydrogen) plasma treatment is performed for the porous SiO₂ film 107.

[0091] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 101 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

[0092] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0093] Subsequently, as shown in FIG. 1F, an SiO₂ film 108 is formed on the porous SiO₂ film 107. This SiO₂ film 108 is formed by a CVD method, which employs source gas containing TEOS and O₃. In this case, since O₃ contained in source gas has concentration enough for oxidation of TEOS, the SiO₂ film 108 exhibits flowability, and can be planarized to a considerable extent.

[0094] Then, as shown in FIG. 1G, the SiO₂ film 108 is polished by a CMP method (chemical mechanical polishing method) so as to planarize its surface. At this time, some portions of the previously formed SiO₂ films 105 and 107 are eliminated by polishing. The planarizing by the CMP method should be carried out not to eliminate the whole SiO₂ film 105 formed on a convexity 103a of the aluminum wiring layer.

EP 1 037 275 A1

[0095] Subsequently, as shown in FIG. 1H, an SiO₂ film 109 (cover insulating film) is formed on the smoothed surface. This SiO₂ film 109 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. A film thickness of the SiO₂ film 109 is 100 nm.

5 [0096] The foregoing process of forming the SiO₂ films 105 (underlying insulating film), 107 and 109 (cover insulating film) results in formation, on the object 104 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the SiO₂ film 107 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 109 is formed on the porous SiO₂ film 107, incursion of water into the SiO₂ film 107 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 107 can improve the moisture absorption resistance of the film. Moreover, the SiO₂ films 105, 107 and 109 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

10 (Sixth Embodiment)

[0097] A sixth embodiment is a case of applying the fifth embodiment to a damascene process.

15 [0098] FIGS. 2A to 2M are cross-sectional views illustrating the sixth embodiment.

[0099] First, as shown in FIG. 2A, a BPSG (borophosphosilicate glass) film 202 is formed on a silicon substrate 201. Then, after an aluminum layer is formed thereon, patterning is performed for the same to form an aluminum wiring layer 203. It should be noted that the aluminum wiring layer 203 in figures is not patterned for convenience. The silicon substrate 201, the BPSG film 202 and the aluminum wiring layer 203 constituted an object 204 to be formed.

20 [0100] As shown in FIG. 2B, an SiO₂ film 205 (underlying insulating film) having a film thickness of 100 nm is formed on the aluminum wiring layer 203. This SiO₂ film 205 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases.

25 [0101] Subsequently, as shown in FIG. 2C, an Si-C-O (-H) film 206 is formed on the SiO₂ film 205 (underlying insulating film). This Si-C-O (-H) film 206 is formed by performing plasma polymerization for TEOS and O₂. This plasma polymerization is performed by applying an RF power having a frequency of 13.56 MHz and power of 100 W to the TEOS and O₂. Flow rates of source gases at this time are 30 sccm and 240 sccm for TEOS and O₂, respectively. In forming the Si-C-O (-H) film 206 the temperature of the silicon substrate 201 is maintained at 400 °C and the pressure is held at 1 Torr. A film thickness of the Si-C-O (-H) film 206 is 500 nm.

30 [0102] It should be noted that, in forming the Si-C-O (-H) 206, one of methylsilane (Si(CH₃)H₃) and trimethylsilylborate ([(CH₃)₃SiO]₃B) can also be used instead of the TEOS. When using one of these gases, the flow rate of the gas is 30 sccm and other process conditions are the same as in the case using the TEOS. Using the trimethylsilylborate, the Si-C-O (-H) film 206 further contains B (boron).

35 [0103] Then, as shown in FIG. 2D, O (oxygen) plasma treatment is performed for the Si-C-O (-H) film 206. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 201 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

40 [0104] In this O (oxygen) plasma treatment C or H contained in the Si-C-O (-H) film 206 is oxidized and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C-O (-H) film 206 becomes a porous SiO₂ film 207.

[0105] Subsequently, as shown in FIG. 2E, H (hydrogen) plasma treatment is performed for the porous SiO₂ film 207.

45 [0106] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 201 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

50 [0107] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0108] Subsequently, as shown in FIG. 2F, patterning is performed for the SiO₂ film 207 to damascene trench 208. This damascene trench 208 reaches the SiO₂ film 205 formed below the SiO₂ film 207.

55 [0109] Then, as shown in FIG. 2G, an SiO₂ film 209 (first insulating film) is formed on the SiO₂ film 207 and on the side and bottom portions of the damascene trench 208. This SiO₂ film 209 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. By the SiO₂ film 209 formed on the side portion of the damascene trench 208, Cu buried later in the damascene trench 208 can be prevented from being dispersed in the porous SiO₂ film 207.

[0110] Then, as shown in FIG. 2H, anisotropic etching is performed for the SiO₂ film 209 (first insulating film) and the SiO₂ film 205 (underlying insulating film) formed below the damascene trench 208. Accordingly, the SiO₂ film 209 is eliminated except for a portion formed on the side portion of the damascene trench 208. Also, since the

EP 1 037 275 A1

SiO₂ film 205 formed below the damascene trench 208 is eliminated, a contact hole that reaches the aluminum wiring layer 203 is formed below the damascene trench 208.

[0111] Subsequently, as shown in FIG. 2I, a Cu-plated film 210 is formed in the damascene trench 208 and on the SiO₂ film 207. The Cu-plated film 210 formed in the damascene trench 208 is used as a Cu wiring line.

5 [0112] Then, as shown in FIG. 2J, the Cu-plated film 210 formed on the SiO₂ film 207 is polished by a CPM method to be eliminated. Accordingly, the Cu-plated film remains only in the damascene trench 208.

[0113] Subsequently, as shown in FIG. 2K, a barrier metal TiN film 211 is formed above the damascene trench 208. Accordingly, Cu in the damascene trench 208 can be prevented from being dispersed in an SiO₂ film later formed above the damascene trench 208.

10 [0114] Then, as shown in FIG. 2L, patterning is performed to leave a TiN film 211a formed above the damascene trench 208, and the TiN film 211 formed in the other portions is etched to be eliminated.

[0115] Subsequently, as shown in FIG. 2M, an SiO₂ film 212 (cover insulating film) is formed on the SiO₂ film 207 and the TiN film 211a. This SiO₂ film 212 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases.

15 [0116] The foregoing process results in formation, on the object 204 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the SiO₂ film 207 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 212 (cover insulating film) is formed on the porous SiO₂ film 207, incursion of water into the SiO₂ film 207 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 207 can improve the moisture absorption resistance of the film. Moreover, the SiO₂ films 207 and 212 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

20 (Seventh Embodiment)

[0117] A seventh embodiment is different from the first to sixth embodiments in that, instead of forming a porous SiO₂ film, a porous B-containing SiO₂ film is formed.

25 [0118] FIGS. 1A to 1H are cross-sectional views illustrating the seventh embodiment.

[0119] First, as shown in FIG. 1A, a BPSG (borophosphosilicate glass) film 102 is formed on a silicon substrate 101. Then, after an aluminum film is formed on the BPSG film 102, patterning is performed for the same to form an aluminum wiring layer 103. The silicon substrate 101, the BPSG film 102 and the aluminum wiring layer 103 formed in this manner constitute an object 104 to be formed.

30 [0120] Then, as shown in FIG. 1B, an SiO₂ film 105 (underlying insulating film) is formed on the object 104 to be formed. This SiO₂ film 105 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases. A film thickness of the SiO₂ film 105 is 100 nm. The SiO₂ film 105 can prevent H₂O from being dispersed in the aluminum wiring layer 103, which may cause corrosion.

35 [0121] Subsequently, as shown in FIG. 1C, a carbon and boron, or carbon, boron and hydrogen containing (referred to as Si-C-B (-H), hereinafter) film 106 is formed on the SiO₂ film 105 (underlying insulating film). This Si-C-B (-H) film 106 is formed by using TEOS and B₂H₆ as source gases, applying an RF power having a frequency of 13.56 MHz and a power of 100 W while keeping the silicon substrate 101 at 100 °C, and performing plasma polymerization for TEOS and B₂H₆ at pressure of 1 Torr. Flow rates of source gases at this time are 30 sccm and 24 sccm for TEOS and B₂H₆ respectively. A film thickness of this Si-C-B (-H) film 106 is 500 nm.

40 [0122] It should be noted that, in forming the Si-C-B (-H) film 106, one of methylsilane (Si(CH₃)₃) and trimethylsilylborate ([(CH₃)₃SiO]₃B) can also be used instead of the TEOS. When using one of these gases, the flow rate of the gas is 30 sccm and the other process conditions are the same as in the case using the TEOS.

[0123] Furthermore, one of TMB (B(OC₂H₅)₃) and TEB (B(OC₂H₅)₃) can also be used instead of the B₂H₆. When using one of these gases, the flow rate of the gas is 48 sccm and the other process condition is the same as in the above.

45 [0124] Then, as shown in FIG. 1D, O (oxygen) plasma treatment is performed for the Si-C-B (-H) film 106. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 101 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

50 [0125] In this O (oxygen) plasma treatment C or H contained in the Si-C-B (-H) film 106 is oxidized and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C-B (-H) film 106 becomes a porous B-containing SiO₂ film 107.

[0126] Subsequently, as shown in FIG. 1E, H (hydrogen) plasma treatment is performed for the porous B-containing SiO₂ film 107.

[0127] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 101 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

EP 1 037 275 A1

[0128] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0129] Subsequently, as shown in FIG. 1F, an SiO₂ film 108 is formed on the porous B-containing SiO₂ film 107. This SiO₂ film 108 is formed by a CVD method, which contains TEOS and O₃ as source gases. In this case, since O₃ in source gas has concentration enough for oxidation of TEOS, the SiO₂ film 108 exhibits flowability, and can be planarized to a considerable extent.

[0130] Then, as shown in FIG. 1G, the SiO₂ film 108 is polished by a CMP method (chemical mechanical polishing method) to planarize its surface. At this time, some portions of the SiO₂ film 105 and the B-containing SiO₂ film 107 which have been formed before are eliminated by polishing. The planarizing by the CMP method should be carried out not to eliminate the whole SiO₂ film 105 formed on a convexity 103a of the aluminum wiring layer.

[0131] Subsequently, as shown in FIG. 1H, an SiO₂ film 109 (cover insulating film) is formed on the planarized surface. This SiO₂ film 109 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. A film thickness of the SiO₂ film 109 is 100 nm.

[0132] The foregoing process of forming the SiO₂ films 105 (underlying insulating film), 109 (cover insulating film) and the B-containing SiO₂ film 107 results in formation, on the object 104 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the B-containing SiO₂ film 107 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 109 is formed on the porous B-containing SiO₂ film 107, incursion of water into the B-containing SiO₂ film 107 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 107 can improve the moisture absorption resistance of the film. Moreover, the SiO₂ films 105 and 109 and the B-containing SiO₂ film 107 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

(Eighth Embodiment)

[0133] An eighth embodiment is a case of applying the seventh embodiment to a damascene process.

[0134] FIGS. 2A to 2M are cross-sectional views illustrating the eighth embodiment.

[0135] First, as shown in FIG. 2A, a BPSG (borophosphosilicate glass) film 202 is formed on a silicon substrate 201. Then, after an aluminum layer is formed thereon, patterning is performed for the same to form an aluminum wiring layer 203. It should be noted that the aluminum wiring layer 203 in figures is not patterned for convenience. The silicon substrate 201, the BPSG film 202 and the aluminum wiring layer 203 constitute an object 204 to be formed.

[0136] As shown in FIG. 2B, an SiO₂ film 205 (underlying insulating film) having a film thickness of 100 nm is formed on the aluminum wiring layer 203. This SiO₂ film 205 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases.

[0137] Subsequently, as shown in FIG. 2C, an Si-C-B (-H) film 206 is formed on the SiO₂ film 205 (underlying insulating film). This Si-C-B (-H) film 206 is formed by using TEOS (Tetra-Ethyl-Ortho-Silicate) and B₂H₆ as source gases, applying an RF power having a frequency of 13.56 MHz and a power of 100 W while keeping the silicon substrate 201 at 100°C, and performing plasma polymerization for TEOS and B₂H₆ at pressure of 1 Torr. At this time, flow rates of source gases are 30 sccm and 24 sccm for TEOS and B₂H₆ respectively. A film thickness of the Si-C-B (-H) film 206 is 500 nm.

[0138] It should be noted that, in forming the Si-C-B (-H) film 206, one of methylsilane (Si(CH₃)H₃) and trimethylsilylborate (((CH₃)₃SiO)₃B) can also be used instead of the TEOS. When using one of these gases, the flow rate of the gas is 30 sccm and the other process conditions are the same as in the case using the TEOS.

[0139] Furthermore, one of TMB (B(OCH₃)₃) and TEB (B(OC₂H₅)₃) can also be used instead of the B₂H₆. When using one of these gases, the flow rate of the gas is 48 sccm and the other process condition is the same as in the above.

[0140] Then, as shown in FIG. 2D, O (oxygen) plasma treatment is performed for the Si-C-B (-H) film 206. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 201 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0141] In this O (oxygen) plasma treatment C or H contained in the Si-C-B (-H) film 206 is oxidized and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed. Accordingly, the Si-C-B (-H) film 206 becomes a porous B-containing SiO₂ film 207.

[0142] Subsequently, as shown in FIG. 2E, H (hydrogen) plasma treatment is performed for the porous B-containing SiO₂ film 207.

[0143] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 201 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

EP 1 037 275 A1

[0144] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0145] Subsequently, as shown in FIG. 2F, patterning is performed for the B-containing SiO₂ film 207 to form a damascene trench 208. This damascene trench 208 reaches the SiO₂ film 205 formed below the B-containing SiO₂ film 207.

[0146] Then, as shown in FIG. 2G, an SiO₂ film 209 (first insulating film) is formed on the B-containing SiO₂ film 207 and on the side and bottom portions of the damascene trench 208. This SiO₂ film 209 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. By the SiO₂ film 209 formed on the side portion of the damascene trench 208, Cu later buried in the damascene trench 208 can be prevented from being dispersed in the porous B-containing SiO₂ film 207.

[0147] Then, as shown in FIG. 2H, anisotropic etching is performed for the SiO₂ film 209 (first insulating film) and the SiO₂ film 205 (underlying insulating film) formed below the damascene trench 208. Accordingly, the SiO₂ film 209 is eliminated except for a portion formed on the side portion of the damascene trench 208. Also, since the SiO₂ film 205 formed below the damascene trench 208 is eliminated, a contact hole that reaches the aluminum wiring layer 203 is formed below the damascene trench 208.

[0148] Subsequently, as shown in FIG. 2I, a Cu-plated film 210 is formed in the damascene trench 208 and on the B-containing SiO₂ film 207. The Cu-plated film 210 formed in the damascene trench 208 is used as a Cu wiring line.

[0149] Then, as shown in FIG. 2J, the Cu-plated film 210 formed on the B-containing SiO₂ film 207 is polished by a CMP method to be eliminated. Accordingly, the Cu-plated film 210 remains only in the damascene trench 208.

[0150] Subsequently, as shown in FIG. 2K, a barrier metal TiN film 211 is formed above the damascene trench 208. Accordingly, Cu in the damascene trench 208 can be prevented from being dispersed in an SiO₂ film formed later above the damascene trench 208.

[0151] Then, as shown in FIG. 2L, patterning is performed to leave a TiN film 211a formed above the damascene trench 208, and the TiN film 211 formed on the other portions are etched to be eliminated.

[0152] Subsequently, as shown in FIG. 2M, an SiO₂ film 212 (cover insulating film) is formed on the B-containing SiO₂ film 207 and the TiN film 211a. This SiO₂ film 212 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases.

[0153] The foregoing process results in formation, an the object 204 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the B-containing SiO₂ film 207 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 212 (cover insulating film) is formed on the porous B-containing SiO₂ film 207, incursion of water into the B-containing SiO₂ film 207 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 207 can improve the moisture absorption resistance of the film. Moreover, the B-containing SiO₂ film 207 and the SiO₂ film 212 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

[0154] In a ninth embodiment, a film containing carbon, boron and oxygen, or a film containing carbon, boron, oxygen and hydrogen (referred to as Si-C-B-O (-H) film, hereinafter) is formed in order to form a porous B-containing SiO₂ film.

[0155] FIGS. 1A to 1H are cross-sectional views illustrating the ninth embodiment.

[0156] First, as shown in FIG. 1A, a BPSG (borophosphosilicate glass) film 102 is formed on a silicon substrate 101. Then, after an aluminum film is formed on the BPSG film 102, patterning is performed for the same to form an aluminum wiring layer 103. The silicon substrate 101, the BPSG film 102 and the aluminum wiring layer 103 formed in this manner constitute an object 104 to be formed.

[0157] Then, as shown in FIG. 1B, an SiO₂ film 105 (underlying insulating film) is formed on the object 104 to be formed. This SiO₂ film 105 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases. A film thickness of the SiO₂ film 105 is 100 nm. The SiO₂ film 105 can prevent H₂O from being dispersed in the aluminum wiring layer 103, which may cause corrosion.

[0158] Subsequently, as shown in FIG. 1C, an Si-C-B-O (-H) film 106 is formed on the SiO₂ film 105 (underlying insulating film). This Si-C-B-O (-H) film 106 is formed by performing plasma polymerization for TEOS, B₂H₆ and O₂. This plasma polymerization is performed by applying an RF power having frequency of 13.56 MHz and power of 100 W to the TEOS, B₂H₆ and O₂. And the flow rates of source gases at this time are 30 sccm, 24 sccm and 260 sccm for TEOS, B₂H₆ and O₂ respectively. In forming the Si-C-B-O (-H) film 106 the temperature of the silicon substrate 101 is maintained at 400 °C and the pressure is held at 1 Torr. A film thickness of the Si-C-B-O (-H) film 106 is 500 nm.

[0159] It should be noted that, in forming the Si-C-B-O (-H) film 106, one of methylsilane (Si(CH₃)₃) and trimethylsilylborate (((CH₃)₃SiO)₃B) can also be used instead of the TEOS. When using one of these gases, the flow rate of the gas is 30 sccm and the other process conditions are the same as in the case using the TEOS.

[0160] Furthermore, one of TMB (B(OCH₃)₃) and TEB (B(OC₂H₅)₃) can also be used instead of the B₂H₆. When using one of these gases, the flow rate of the gas is 48 sccm and the other process conditions are the same as in the above.

EP 1 037 275 A1

[0161] Then, as shown in FIG. 1D, O (oxygen) plasma treatment is performed for the Si-C-B-O (-H) film 106. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 101 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0162] In this O (oxygen) plasma treatment C or H contained in the Si-C-B-O (-H) film 106 is oxidized, and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C-B-O (-H) film 106 becomes a porous B-containing SiO₂ film 107.

[0163] Subsequently, as shown in FIG. 1E, H (hydrogen) plasma treatment is performed for the porous B-containing SiO₂ film 107.

[0164] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 101 is maintained at 400 °C. Still, further, the time for the H plasma treatment is 60 sec.

[0165] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0166] Subsequently, as shown in FIG. 1F, an SiO₂ film 108 is formed on the porous B-containing SiO₂ film 107. This SiO₂ film 108 is formed by a CVD method, which contains TEOS and O₃ as source gases. In this case, since O₃ in source gas has concentration enough for oxidation of TEOS, the SiO₂ film 108 exhibits flowability, and can be planarized to a considerable extent.

[0167] Then, as shown in FIG. 1G, the SiO₂ film 108 is polished by a CMP method (chemical mechanical polishing method) to planarize its surface. At this time, some portions of the SiO₂ film 105 and the B-containing SiO₂ film 107 which have been formed before are eliminated by polishing. The planarizing by the CMP method should be carried out not to eliminate the whole SiO₂ film 105 formed on a convexity 103a of the aluminum wiring layer.

[0168] Subsequently, as shown in FIG. 1H, an SiO₂ film 109 (cover insulating film) is formed on the planarized surface. This SiO₂ film 109 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. A film thickness of the SiO₂ film 109 is 100 nm.

[0169] The foregoing process of forming the SiO₂ films 105 (underlying insulating film) and 109 (cover insulating film), and the B-containing SiO₂ film 107 results in formation, on the object 104 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the B-containing SiO₂ film 107 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 109 is formed on the porous B-containing SiO₂ film 107, incursion of water into the B-containing SiO₂ film 107 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 107 can improve the moisture absorption resistance of the film. Moreover, the SiO₂ films 105 and 109 and the B-containing SiO₂ film 107 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

(Tenth Embodiment)

[0170] A tenth embodiment is a case of applying the ninth embodiment to a damascene process.

[0171] FIGS. 2A to 2M are cross-sectional views illustrating the tenth embodiment.

[0172] First, as shown in FIG. 2A, a BPSG (borophosphosilicate glass) film 202 is formed on a silicon substrate 201. After an aluminum layer is formed thereon, patterning is performed for the same to form an aluminum wiring layer 203. It should be noted that the aluminum wiring layer 203 in figures is not patterned for convenience. The silicon substrate 201, the BPSG film 202 and the aluminum wiring layer 203 constitute an object 204 to be formed.

[0173] Then, as shown in FIG. 2B, an SiO₂ film 205 (underlying insulating film) having a film thickness of 100 nm is formed on the aluminum wiring layer 203. This SiO₂ film 205 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases.

[0174] Subsequently, as shown in FIG. 2C, an Si-C-B-O (-H) film 206 is formed on the SiO₂ film 205 (underlying insulating film). This Si-C-B-O (-H) film 206 is formed by using TEOS (Tetra-Ethyl-Ortho-Silicate), B₂H₆ and O₂ as source gases, and performing plasma polymerization for these source gases. This plasma polymerization is performed by applying an RF power having frequency of 13.56 MHz and power of 100 W to the TEOS, B₂H₆ and O₂. And the flow rates of source gases at this time are 30 sccm, 24 sccm and 260 sccm for TEOS, B₂H₆ and O₂ respectively. In forming the Si-C-B-O (-H) film 206 the temperature of the silicon substrate 201 is maintained at 400 °C and the pressure is held at 1 Torr. A film thickness of the Si-C-B-O (-H) film 206 is 500 nm.

[0175] It should be noted that, in forming the Si-C-B-O (-H) film 206, one of methylsilane (Si(CH₃)H₃) and trimethylsilylborate (((CH₃)₃SiO)₃B) can also be used instead of the TEOS. When using one of these gases, the flow rate of the gas is 30 sccm and the other process conditions are the same as in the case using the TEOS.

[0176] Furthermore, one of TMB (B(OCH₃)₃) and TEB (B(OC₂H₅)₃) can also be used instead of the B₂H₆. When using one of these gases, the flow rate of the gas is 48 sccm and the other process conditions are the same as in

the above.

[0177] Then, as shown in FIG. 2D, O (oxygen) plasma treatment is performed for the Si-C-B-O (-H) film 206. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 201 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0178] In this O (oxygen) plasma treatment C or H contained in the Si-C-B-O (-H) film 206 is oxidized, and discharged to the outside. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C-B-O (-H) film 206 becomes a porous B-containing SiO₂ film 207.

[0179] Subsequently, as shown in FIG. 2E, H (hydrogen) plasma treatment is performed for the porous B-containing SiO₂ film 207.

[0180] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400 W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 201 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

[0181] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0182] Subsequently, as shown in FIG. 2F, patterning is performed for the B-containing SiO₂ film 207 to form a damascene trench 208. This damascene trench 208 reaches the SiO₂ film 205 formed below the B-containing SiO₂ film 207.

[0183] Then, as shown in FIG. 2G, an SiO₂ film 209 (first insulating film) is formed on the B-containing SiO₂ film 207 and on the side and bottom portions of the damascene trench 208. This SiO₂ film 209 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. By the SiO₂ film 209 formed on the side portion of the damascene trench 208, Cu buried later in the damascene trench 208 can be prevented from being dispersed in the porous B-containing SiO₂ film 207.

[0184] Then, as shown in FIG. 2H, anisotropic etching is performed for the SiO₂ film 209 (first insulating film) and the SiO₂ film 205 (underlying insulating film) formed below the damascene trench 208. Accordingly, the SiO₂ film 209 is eliminated except for a portion formed on the side portion of the damascene trench 208. Also, since the SiO₂ film 205 formed below the damascene trench 208 is eliminated, a contact hole that reaches the aluminum wiring layer 203 is formed below the damascene trench 208.

[0185] Subsequently, as shown in FIG. 2I, a Cu-plated film 210 is formed in the damascene trench 208 and on the B-containing SiO₂ film 207. The Cu-plated film 210 formed in the damascene trench 208 is used as a Cu wiring line.

[0186] Then, as shown in FIG. 2J, the Cu-plated film 210 formed on the B-containing SiO₂ film 207 is polished by a CMP method to be eliminated. Accordingly, the Cu-plated film remains only in the damascene trench 208.

[0187] Subsequently, as shown in FIG. 2K, a barrier metal TiN film 211 is formed above the damascene trench 208. Accordingly, Cu in the damascene trench 208 can be prevented from being dispersed in an SiO₂ film formed later above the damascene trench 208.

[0188] Then, as shown in FIG. 2L, patterning is performed to leave a TiN film 211a formed above the damascene trench 208, and the TiN film 211 formed on the other portions are etched to be eliminated.

[0189] Subsequently, as shown in FIG. 2M, an SiO₂ film 212 (cover insulating film) is formed on the B-containing SiO₂ film 207 and the TiN film 211a. This SiO₂ film 212 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases.

[0190] The foregoing process results in formation, on the object 204 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the B-containing SiO₂ film 207 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 212 (cover insulating film) is formed on the porous B-containing SiO₂ film 207, incursion of water into the B-containing SiO₂ film 207 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 207 can improve the moisture absorption resistance of the film. Moreover, the B-containing SiO₂ film 207 and the SiO₂ film 212 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

50 (Eleventh Embodiment)

[0191] An eleventh embodiment is different from the first to tenth embodiments in that a porous F-containing SiO₂ film is formed instead of forming a porous SiO₂ film or a porous B-containing SiO₂ film.

[0192] FIGS. 1A to 1H are cross-sectional views illustrating the eleventh embodiment.

[0193] First, as shown in FIG. 1A, a BPSG (borophosphosilicate glass) film 102 is formed on a silicon substrate 101. Then, after an aluminum film is formed on the BPSG film 102, patterning is performed for the same to form an aluminum wiring layer 103. The silicon substrate 101, the BPSG film 102 and the aluminum wiring layer 103 constitute an object 104 to be formed.

EP 1 037 275 A1

[0194] Then, as shown in FIG. 1B, an SiO₂ film 105 (underlying insulating film) is formed on the object 104 to be formed. This SiO₂ film 105 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH₄ and N₂O are used as source gases. A film thickness of the SiO₂ film 105 is 100 nm. The SiO₂ film 105 can prevent H₂O from being dispersed in the aluminum wiring layer 103, which may cause corrosion.

5 [0195] Subsequently, as shown in FIG. 1C, a film 106 containing carbon and fluorine or carbon, fluorine and hydrogen (referred to as Si-C-F (-H) film, hereinafter) is formed on the SiO₂ film 105 (underlying insulating film). This Si-C-F (-H) film 106 is formed by using TEOS and C₂F₆ as source gases, applying an RF power having a frequency of 13.56 MHz and a power of 100 W while keeping the silicon substrate 101 at 100 °C, and performing plasma polymerization for TEOS and C₂F₆ at pressure of 1 Torr. Flow rates of source gases at this time are 30 sccm and 5 sccm for TEOS and C₂F₆ respectively. A film thickness of the Si-C-F (-H) film 106 is 500 nm.

10 [0196] It should be noted that in forming the Si-C-F (-H) film 106, metylsilane (Si(CH₃)H₃) can also be used instead of the TEOS. When using the metylsilane, its flow rate is 30 sccm and the other process conditions are the same as in the case using the TEOS.

15 [0197] Then, as shown in FIG. 1D, O (oxygen) plasma treatment is performed for the Si-C-F (-H) film 106. This O (oxygen) plasma treatment is performed supplying 600 sccm O₂ to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O₂. The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 101 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

20 [0198] In this O (oxygen) plasma treatment C or H contained in the Si-C-F (-H) film 106 is oxidized, and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C-F (-H) film 106 becomes a porous F-containing SiO₂ film 107.

25 [0199] Subsequently, as shown in FIG. 1E, H (hydrogen) plasma treatment is performed for the porous F-containing SiO₂ film 107.

[0200] This H plasma treatment is performed by supplying H₂ of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50 W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 101 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

30 [0201] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

35 [0202] Subsequently, as shown in FIG. 1F, an SiO₂ film 108 is formed on the porous F-containing SiO₂ film 107. This SiO₂ film 108 is formed by a CVD method, which contains TEOS and O₃ as source gases. In this case, since O₃ in source gas had concentration enough for oxidation of TEOS, the SiO₂ film 108 exhibits flowability, and can be planarized to a considerable extent.

40 [0203] Then, as shown in FIG. 1G, the SiO₂ film 108 is polished by a CMP method (chemical mechanical polishing method) to planarize its surface. At this time, some portions of the SiO₂ film 105 and the F-containing SiO₂ film 107 which have been formed before are eliminated by polishing. The planarizing by the CMP method should be carried out not to eliminate the whole SiO₂ film 105 formed in a convexity 103a of the aluminum wiring layer.

45 [0204] Subsequently, as shown in FIG. 1H, an SiO₂ film 109 (cover insulating film) is formed on the planarized surface. This SiO₂ film 109 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases. A film thickness of the SiO₂ film 109 is 100 nm.

50 [0205] The foregoing process of forming the SiO₂ films 105 (underlying insulating film) and 109 (cover insulating film), and the F-containing SiO₂ film 107 results in formation, on the object 104 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the F-containing SiO₂ film 107 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 109 is formed on the porous F-containing SiO₂ film 107, incursion of water into the F-containing SiO₂ film 107 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 107 can improve the moisture absorption resistance of the film. Moreover, the SiO₂ films 105 and 109, and the F-containing SiO₂ film 107 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

55 (Twelfth Embodiment)

[0206] A twelfth embodiment is a case of applying the eleventh embodiment to a damascene process.

[0207] FIGS. 2A to 2M are cross-sectional views illustrating the twelfth embodiment.

55 [0208] First, as shown in FIG. 2A, a BPSG (borophosphosilicate glass) film 202 is formed on a silicon substrate 201. Then, after an aluminum layer is formed thereon, patterning is performed for the same to form an aluminum wiring layer 203. It should be noted that the aluminum wiring layer 203 in figures is not patterned for convenience. The silicon substrate 201, the BPSG film 202 and the aluminum wiring layer 203 constituted an object 204 to be formed.

[0209] As shown in FIG. 2B, an SiO₂ film 205 (underlying insulating film) having a film thickness of 100 nm is

EP 1 037 275 A1

formed on the aluminum wiring layer 203. This SiO_2 film 205 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH_4 and N_2O are used as source gases.

5 [0210] Subsequently, as shown in FIG. 2C, an Si-C-F (-H) film 206 is formed on the SiO_2 film 205 (underlying insulating film). This Si-C-F (-H) film 206 is formed by using TEOS (Tetra-Ethyl-Ortho-Silicate) and C_2F_6 as source gases, applying an RF power having a frequency of 13.56 MHz and a power of 100 W while keeping the silicon substrate 201 at 100 °C, and performing plasma polymerization for TEOS and C_2F_6 at pressure of 1 Torr. Flow rates of source gases at this time are 30 sccm and 5 sccm for TEOS and C_2F_6 respectively. A film thickness of the Si-C-F (-H) film 206 is 500 nm.

10 [0211] It should be noted that in forming the Si-C-F (-H) film 206, methylsilane ($\text{Si}(\text{CH}_3)_\text{H}_3$) can also be used instead of the TEOS. When using the methylsilane, its flow rate is 30 sccm and the other process conditions are the same as in the case using the TEOS.

15 [0212] Then, as shown in FIG. 2D, O (oxygen) plasma treatment is performed for the Si-C-F (-H) film 206. This O (oxygen) plasma treatment is performed supplying 600 sccm O_2 to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O_2 . The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 201 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

20 [0213] In this O (oxygen) plasma treatment C or H contained in the Si-C-F (-H) film 206 is oxidized, and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C-F (-H) film 206 becomes a porous F-containing SiO_2 film 207.

25 [0214] Subsequently, as shown in FIG. 2E, H (hydrogen) plasma treatment is performed for the porous F-containing SiO_2 film 207.

30 [0215] This H plasma treatment is performed by supplying H_2 of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 201 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

35 [0216] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

40 [0217] Subsequently, as shown in FIG. 2F, patterning is performed for the F-containing SiO_2 film 207 to form a damascene trench 208. This damascene trench 208 reaches the SiO_2 film 205 formed below the F-containing SiO_2 film 207.

45 [0218] Then, as shown in FIG. 2G, an SiO_2 film 209 (first insulating film) is toned on the F-containing SiO_2 film 207 and on the side and bottom portions of the damascene trench 208. This SiO_2 film 209 is formed by a plasma enhanced CVD method, and SiH_4 and N_2O are used as source gases. By the SiO_2 film 209 formed on the side portion of the damascene trench 208, Cu buried later in the damascene trench 208 can be prevented from being dispersed in the porous F-containing SiO_2 film 207.

50 [0219] Then, as shown in FIG. 2H, anisotropic etching is performed for the SiO_2 film 209 (first insulating film) and the SiO_2 film 205 (underlying insulating film) formed below the damascene trench 208. Accordingly, the SiO_2 film 209 is eliminated except for a portion formed on the side portion of the damascene trench 208. Also, since the SiO_2 film 205 formed below the damascene trench 208 is eliminated, a contact hole that reaches the aluminum wiring layer 203 is formed below the damascene trench 208.

55 [0220] Subsequently, as shown in FIG. 2I, a Cu-plated film 210 is formed in the damascene trench 208 and on the F-containing SiO_2 film 207. The Cu-plated film 210 formed in the damascene trench 208 is used as a Cu wiring line.

60 [0221] Then, as shown in FIG. 2J, the Cu-plated film 210 formed on the F-containing SiO_2 film 207 is polished by a CMP method to be eliminated. Accordingly, the Cu-plated film remains only in the damascene trench 208.

65 [0222] Subsequently, as shown in FIG. 2K, a barrier TiN film 211 is formed above the damascene trench 208. Accordingly, Cu in the damascene trench 208 can be prevented from being dispersed in an SiO_2 film formed later above the damascene trench 208.

70 [0223] Then, as shown in FIG. 2L, patterning is performed to leave a TiN film 211a formed above the damascene trench 208, and the TiN film 211 formed in the other portions is etched to be eliminated.

75 [0224] Subsequently, as shown in FIG. 2M, an SiO_2 film 212 (cover insulating film) is formed on the F-containing SiO_2 film 207 and the TiN film 211a. This SiO_2 film 212 is formed by a plasma enhanced CVD method, and SiH_4 and N_2O are used as source gases.

80 [0225] The foregoing process results in formation, on the object 204 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the F-containing SiO_2 film 207 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO_2 film. Also, since the usual SiO_2 film 212 (cover insulating film) is formed on the porous F-containing SiO_2 film 207, incursion of water into the F-containing SiO_2 film 207 can be prevented. Furthermore, performing the H plasma treatment for the SiO_2 film 207 can improve the moisture absorption resistance of the film. Moreover, the F-containing SiO_2 film 207 and the SiO_2 film 212 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

(Thirteenth Embodiment)

[0226] In a thirteenth embodiment, a film containing carbon, fluorine and oxygen, or a film containing carbon, fluorine, oxygen and hydrogen (referred to as Si-C-F-O (-H) film, hereinafter) is formed in order to form a porous F-containing SiO_2 film.

[0227] FIGS. 1A to 1H are cross-sectional views illustrating the thirteenth embodiment.

[0228] First, as shown in FIG. 1A, a BPSG (borophosphosilicate glass) film 102 is formed on a silicon substrate 101. Then, after an aluminum film is formed on the BPSG film 102, patterning is performed for the same to form an aluminum wiring layer 103. The silicon substrate 101, the BPSG film 102 and the aluminum wiring layer 103 formed in this manner constitute an object 104 to be formed.

[0229] Then, as shown in FIG. 1B, an SiO_2 film 105 (underlying insulating film) is formed on the object 104 to be formed. This SiO_2 film 105 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH_4 and N_2O are used as source gases. A film thickness of the SiO_2 film 105 is 100 nm. The SiO_2 film 105 can prevent H_2O from being dispersed in the aluminum wiring layer 103, which may cause corrosion.

[0230] Subsequently, as shown in FIG. 1C, an Si-C-F-O (-H) film 106 is formed on the SiO_2 film 105 (underlying insulating film). This Si-C-F-O (-H) film 106 is formed by performing plasma polymerization for TEOS, C_2F_6 and O_2 . This plasma polymerization is performed by applying an RF power having frequency of 13.56 MHz and power of 100 W to the TEOS, C_2F_6 and O_2 . And the flow rates of source gases at this time are 30 sccm, 5 sccm and 260 sccm for TEOS, C_2F_6 and O_2 respectively.

[0231] In forming the Si-C-F-O (-H) film 106 the temperature of the silicon substrate 101 is maintained at 400 °C and the pressure is held at 1 Torr. A film thickness of the Si-C-F-O (-H) film 106 is 500 nm.

[0232] It should be noted that in forming the Si-C-F-O (-H) film 106, methylsilane ($\text{Si}(\text{CH}_3)_3$) can also be used instead of the TEOS. When using the methylsilane, its flow rate is 30 sccm and the other process conditions are the same as in the case using the TEOS.

[0233] Then, as shown in FIG. 1D, O (oxygen) plasma treatment is performed for the Si-C-F-O (-H) film 106. This O (oxygen) plasma treatment is performed supplying 600 sccm O_2 to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O_2 . The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 101 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0234] In this O (oxygen) plasma treatment C or H contained in the Si-C-F-O (-H) film 106 is oxidized and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C-F-O (-H) film 106 becomes a porous F-containing SiO_2 film 107.

[0235] Subsequently, as shown in FIG. 1E, H (hydrogen) plasma treatment is performed for the porous F-containing SiO_2 film 107.

[0236] This H plasma treatment is performed by supplying H_2 of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 101 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

[0237] The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0238] Accordingly, Si-H bond is substituted for Si dangling bond in the Si-O bond on the surface of the void, and moisture absorption resistance of the film can be increased.

[0239] Subsequently, as shown in FIG. 1F, an SiO_2 film 108 is formed on the porous F-containing SiO_2 film 107. This SiO_2 film 108 is formed by a CVD method, which contains TEOS and O_3 as source gases. In this case, since O_3 in source gas has concentration enough for oxidation of TEOS, the SiO_2 film 108 exhibits flowability, and can be planarized to a considerable extent.

[0240] Then, as shown in FIG. 1G, the SiO_2 film 108 is polished by a CMP method (chemical mechanical polishing method) to planarize its surface. At this time, some portions of the SiO_2 film 105 and the F-containing SiO_2 film 107 which have been formed before are eliminated by polishing. The planarizing by the CMP method should be carried out not to eliminate the whole SiO_2 film 105 formed on a convexity 103a of the aluminum wiring layer.

[0241] Subsequently, as shown in FIG. 1H, an SiO_2 film (cover insulating film) is formed on the planarized surface. This SiO_2 film 109 is formed by a plasma enhanced CVD method, and SiH_4 and N_2O are used as source gases. A film thickness of the SiO_2 film 109 is 100 nm.

[0242] The foregoing process of forming the SiO_2 films 105 (underlying insulating film) and 109 (cover insulating film), and the F-containing SiO_2 film 107 results in formation, on the object 104 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the F-containing SiO_2 film 107 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO_2 film. Also, since the usual SiO_2 film 109 is formed on the porous F-containing SiO_2 film 107, incursion of water into the F-containing SiO_2 film 107 can be prevented. Furthermore, performing the H

EP 1 037 275 A1

plasma treatment for the SiO_2 film 107 can improve the moisture absorption resistance of the film. Moreover, the SiO_2 films 105 and 109 and the F-containing SiO_2 film 107 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

5 (Fourteenth Embodiment)

[0243] A fourteenth embodiment is a case of applying the thirteenth embodiment to a damascene process.

[0244] FIGS. 2A to 2M are cross-sectional views illustrating the fourteenth embodiment.

[0245] 10 First, as shown in FIG. 2A, a BPSG (borophosphosilicate glass) film 202 is formed on a silicon substrate 201. Then, after an aluminum layer is formed thereon, patterning is performed for the same to form an aluminum wiring layer 203. It should be noted that the aluminum wiring layer 203 in figures is not patterned for convenience. The silicon substrate 201, the BPSG film 202 and the aluminum wiring layer 203 constitute an object 204 to be formed.

[0246] 15 As shown in FIG. 2B, an SiO_2 film 205 (underlying insulating film) having a film thickness of 100 nm is formed on the aluminum wiring layer 203. This SiO_2 film 205 is formed by a plasma enhanced CVD method (plasma enhanced chemical vapor deposition method), and SiH_4 and N_2O are used as source gases.

[0247] 20 Subsequently, as shown in FIG. 2C, an Si-C-F-O (-H) film 206 is formed on the SiO_2 film 205 (underlying insulating film). This Si-C-F-O (-H) film 206 is formed by using TEOS (Tetra-Ethyl-Ortho-Silicate), C_2F_6 and O_2 as source gases, and performing plasma polymerization for these source gases. This plasma polymerization is performed by applying an RF power having frequency of 13.56 MHz and power of 100 W to the TEOS, C_2F_6 and O_2 . And the flow rates of source gases at this time are 30 sccm, 5 sccm and 260 sccm for TEOS, C_2F_6 and O_2 respectively.

[0248] 25 In forming the Si-C-F-O (-H) film 206 the temperature of the silicon substrate 201 is maintained at 400 °C and the pressure is held at 1 Torr. A film thickness of the Si-C-F-O (-H) film 206 is 500 nm.

[0249] 30 It should be noted that in forming the Si-C-F-O (-H) film 206, methylsilane ($\text{Si}(\text{CH}_3)_4$) can also be used instead of the TEOS. When using the methylsilane, its flow rate is 30 sccm and the other process conditions are the same as in the case using the TEOS.

[0250] 35 Then, as shown in FIG. 2D, O (oxygen) plasma treatment is performed for the Si-C-F-O (-H) film 206. This O (oxygen) plasma treatment is performed supplying 600 sccm O_2 to a chamber (not shown) and applying RF power having frequency of 400 kHz and power of 300 W to the O_2 . The time for the O (oxygen) plasma treatment is 360 sec, and the temperature of the silicon substrate 201 is maintained at 350 °C during undergoing the O (oxygen) plasma treatment.

[0251] 40 In this O (oxygen) plasma treatment C or H contained in the Si-C-F-O (-H) film 206 is oxidized and discharged to the outside of the film. Voids are formed in portions, from which C or H has been discharged, and Si-O bond is formed on the portion. Accordingly, the Si-C-F-O (-H) film 206 becomes a porous F-containing SiO_2 film 207.

[0252] 45 Subsequently, as shown in FIG. 2E, H (hydrogen) plasma treatment is performed for the porous F-containing SiO_2 film 207.

[0253] 50 This H plasma treatment is performed by supplying H_2 of 600 sccm to a chamber (not shown) and applying RF power to upper and lower electrodes (not shown) that is opposing each other in the chamber. And the RF power applied to the upper electrode has frequency of 13.56 MHz and power of 50W. On the other hand, the RF power applied to the lower electrode has frequency of 400 kHz and power of 400W. Further, during undergoing the H plasma treatment, the pressure in the chamber is 0.1@0.2 Torr and the temperature of the silicon substrate 101 is maintained at 400 °C. Still further, the time for the H plasma treatment is 60 sec.

[0254] 55 The H plasma treatment substitutes Si-H bonds for dangling bonds of Si in an Si-O bond in the surface of the void. Therefore, OH radicals and water are made to be hard to bond to the dangling bonds of Si, which improves the moisture absorption resistance of the film.

[0255] 60 Subsequently, as shown in FIG. 2F, patterning is performed for the B-containing SiO_2 film 207 to form a damascene trench 208. This damascene trench 208 reaches the SiO_2 film 205 formed below the F-containing SiO_2 film 207.

[0256] 65 Then, as shown in FIG. 2G, an SiO_2 film 209 (first insulating film) is formed on the F-containing SiO_2 film 207 and on the side and bottom portions of the damascene trench 208. This SiO_2 film 209 is formed by a plasma enhanced CVD method, and SiH_4 and N_2O are used as source gases. By the SiO_2 film 209 formed on the side portion of the damascene trench 208, Cu buried later in the damascene trench 208 can be prevented from being dispersed in the porous B-containing SiO_2 film 207.

[0257] 70 Then, as shown in FIG. 2H, anisotropic etching is performed for the SiO_2 film 209 and the SiO_2 film 205 (underlying insulating film) formed below the damascene trench 208. Accordingly, the SiO_2 film 209 is eliminated except for a portion formed on the side portion of the damascene trench 208. Also, since the SiO_2 film 205 formed below the damascene trench 208 is eliminated, a contact hole that reaches the aluminum wiring layer 203 is formed below the damascene trench 208.

[0258] 75 Subsequently, as shown in FIG. 2I, a Cu-plated film 210 is formed in the damascene trench 208 and on the B-containing SiO_2 film 207. The Cu-plated film 210 formed in the damascene trench 208 is used as a Cu wiring line.

[0259] 80 Then, as shown in FIG. 2J, the Cu-plated film 210 formed on the F-containing SiO_2 film 207 is polished by a CMP method to be eliminated. Accordingly, the Cu-plated film remains only in the damascene trench 208.

[0260] 85 Subsequently, as shown in FIG. 2K, a barrier metal TiN film 211 is formed above the damascene trench 208. Accordingly, Cu in the damascene trench 208 can be prevented from being dispersed in an SiO_2 film formed

later above the same.

[0261] Then, as shown in FIG. 2L, patterning is performed to leave a TiN film 211a formed above the damascene trench 208, and the TiN film 211 formed on the other portions is etched to be eliminated.

5 [0262] Subsequently, as shown in FIG. 2M, an SiO₂ film 212 (cover insulating film) is formed on the F-containing SiO₂ film 207 and the TiN film 211a. This SiO₂ film 212 is formed by a plasma enhanced CVD method, and SiH₄ and N₂O are used as source gases.

10 [0263] The foregoing process results in formation, on the object 204 to be formed, an interlayer insulating film of a low dielectric constant, which has good heat resistivity and moisture absorption resistance. That is, since the F-containing SiO₂ film 207 has porosity, a relative dielectric constant thereof is smaller than that of a usual SiO₂ film. Also, since the usual SiO₂ film 212 (cover insulating film) is formed on the porous F-containing SiO₂ film 207, incursion of water into the SiO₂ film 207 can be prevented. Furthermore, performing the H plasma treatment for the SiO₂ film 107 can improve the moisture absorption resistance of the film. Moreover, the F-containing SiO₂ film 207 and the SiO₂ film 212 have better heat resistivity compared with the organic insulating film since these films consist mainly of Si and O.

15 **Claims**

1. A method for forming an interlayer insulating film comprising the steps of:

20 applying a RF power to a source gas that contain a compound where said compound contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen), wherein said compound is polymerized by said RF power,

25 said polymerized compound forming a film (106, 206) on an object (104, 204) to be formed where said film (106, 206) contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen); and

making said film (106, 206) be a porous SiO₂ film (107, 207) by oxidizing at least one selected from the group consisting of said C(carbon) and said H(hydrogen) contained in said film (106, 206).

30 2. The method according to claim 1, wherein said source gas further contains O₂ where said O₂ and said compound are polymerized by said RF power to form said film (106, 206) that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen).

35 3. The method according to claim 1, wherein said source gas further contains H₂O where said H₂O and said compound are polymerized by said RF power to form said film (106, 206) that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen).

40 4. The method according to claim 1, wherein said source gas further contains a compound that contains B(boron) and H(hydrogen) where said compound that contains B(boron) and H(hydrogen) are polymerized with said compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen) by said RF power,

wherein said film (106, 206) that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen) further contains B (boron),

45 said film (106, 206) that contain Si(silicon), B(boron), and at least one selected from the group consisting of C(carbon) and H(hydrogen) being consisting of said polymerized compound that contains B(boron) and H(hydrogen) and said polymerized compound that contains Si(silicon) and at least one selected from the group consisting of C (carbon) and H(hydrogen),

50 said porous SiO₂ film (107, 207) further containing B(boron).

55 5. The method according to claim 4, wherein said source gas further contains O₂ where said O₂, said compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H (hydrogen), and said compound that contains B (boron) and H(hydrogen) are polymerized by said RF power, wherein said film (106, 206) that contain Si(silicon), B(boron), and at least one selected from the group consisting of C(carbon) and H(hydrogen) further contains O(oxygen), said film (106, 206) that contains Si(silicon), B(boron), O(oxygen), and at least one selected from the group consisting of C(carbon) and H(hydrogen) being made of said polymerized O₂, said polymerized compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen) and said polymerized compound that contains B (boron) and H (hydrogen).

6. The method according to claim 4, wherein said source gas further contains H₂O where said H₂O, said compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen) and said compound that contains B(boron) and H(hydrogen) are polymerized by said RF power, wherein said film (106,

EP 1 037 275 A1

206) that contains Si(silicon), B(boron), and at least one selected from the group consisting of C(carbon) and H(hydrogen) further contains O(oxygen), said film that contains Si(silicon), B(boron), O(oxygen), and at least one selected from the group consisting of C(carbon) and H(hydrogen) being consisting of said polymerized H₂O, said polymerized compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen) and said polymerized compound that contains B(boron) and H(hydrogen).

5 7. The method according to claim 1, wherein said source gas further contains a compound that contains F(fluorine) and C(carbon) where said compound that contains F(fluorine) and C(carbon) are polymerized with said compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen) by said RF power, wherein said film (106, 206) that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen) further contains F(fluorine), said film that contains Si(silicon), F(fluorine), and at least one selected from the group consisting of C(carbon) and H(hydrogen) being consisting of said polymerized compound that contains F(fluorine) and C(carbon) and said polymerized compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen).

10 8. The method according to claim 7, wherein said source gas further contains O₂ where said O₂, said compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen), and said compound that contains F(fluorine) and C(carbon) are polymerized by said RF power, wherein said film (106, 206) that contains Si(silicon), F(fluorine), and at least one selected from the group consisting of C(carbon) and H(hydrogen) further contains O(oxygen), said film (106, 206) that contains Si(silicon), F(fluorine), O(oxygen), and at least one selected from the group consisting of C(carbon) and H(hydrogen) being consisting of said polymerized O₂, said polymerized compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen), and said polymerized compound that contains C(carbon) and F(fluorine).

15 9. The method according to claim 7, wherein said source gas further contains H₂O where said H₂O, said compound that contains Si (silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen), and said compound that contains F(fluorine) and C(carbon) are polymerized by said RF power, wherein said film (106, 206) that contains Si(silicon), F(fluorine), and at least one selected from the group consisting of C(carbon) and H(hydrogen) further contains O(oxygen), said film (106, 206) that contains Si(silicon), F(fluorine), O(oxygen), and at least one selected from the group consisting of C(carbon) and H(hydrogen) being consisting of said polymerized O₂, said polymerized compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen), and said polymerized compound that contains B(boron) and H(hydrogen).

20 10. The method according to any one of claims 1 to 9, wherein said compound that contains Si(silicon) and at least one selected from the group consisting of C(carbon) and H(hydrogen) is one selected from the group consisting of TEOS, SiH₄, methylsilane (Si(CH₃)₃H₃) and trimethyl-siliruborate((CH₃)₃SiO₃B).

25 11. The method according to any one of claims 4 to 6, wherein said compound that contains B(boron) and H(hydrogen) is one selected from the group consisting of B₂H₆, TMB(B(OCH₃)₃) and TEB(B(OC₂H₅)₃).

30 12. The method according to any one of claims 7 to 9, wherein said compound that contains F(fluorine) and C(carbon) is C₂F₆.

35 13. The method according to any one of claims 1 to 12, wherein an inert gas is added to said source gas.

40 14. The method according to any one of claims 1 to 13, wherein said oxidizing is performed by O(oxygen) plasma.

45 15. The method according to any one of claims 1 to 14, further comprising the steps of:
performing a H(hydrogen) plasma treatment for said porous SiO₂ film (107, 207).

16. The method according to any one of claims 1 to 15, wherein an underlying insulating film (105, 205) is formed on said object (104, 204) to be formed, and said porous SiO₂ film (107, 207) is formed on said underlying insulating film (105, 205).

50 17. The method according to any one of claims 1 to 16, further comprising the steps of:
after forming said porous SiO₂ film (207) on said object (204) to be formed, forming a damascene trench (208) in said porous SiO₂ film (207);
forming a side-wall insulating film on a side portion of said damascene trench (208);
burying a metallic film (210) in said damascene trench (208); and
forming a barrier metal film (211a) on said metallic film (210).

18. The method according to claim 17, wherein said side-wall insulating film is formed by forming a first insulating film (209) on said porous SiO₂ film (207) and on side and bottom portion of said damascene trench (208) after formation of said damascene trench (208), and etching said first insulating film (209) anisotropically to leave a portion thereof formed on said side portion of said damascene trench (208) and remove a portion thereof formed on said bottom portion of said damascene trench (208).

5 19. The method according to any one of claims 1 to 18, further comprising the steps of:

10 forming a cover insulating film (109, 212) on said porous SiO₂ film (107, 207) after forming said porous SiO₂ film (107, 207).

15 20. A semiconductor device employing said porous SiO₂ film (107, 207) where said porous SiO₂ film (107, 207) is formed by method according to any one of claims 1 to 19.

20

25

30

35

40

45

50

55

FIG. 1A

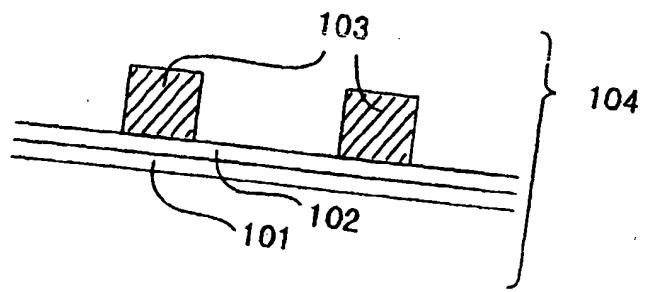


FIG. 1B

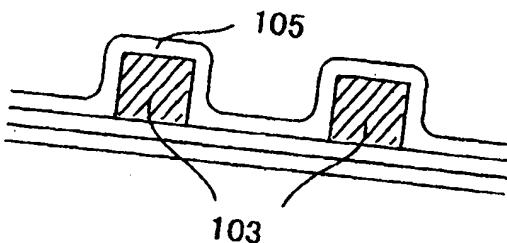


FIG. 1C

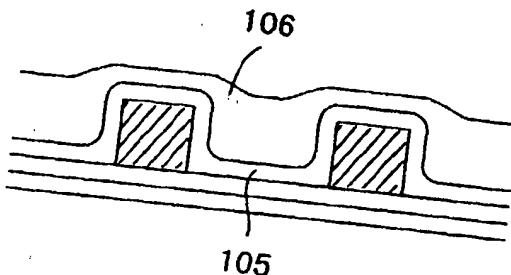
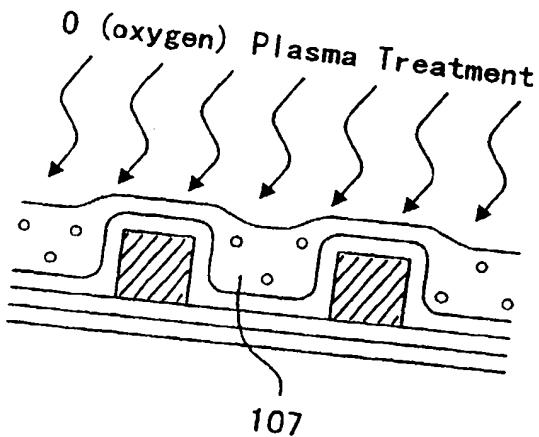


FIG. 1D



H (hydrogen) Plasma Treatment

FIG. 1E

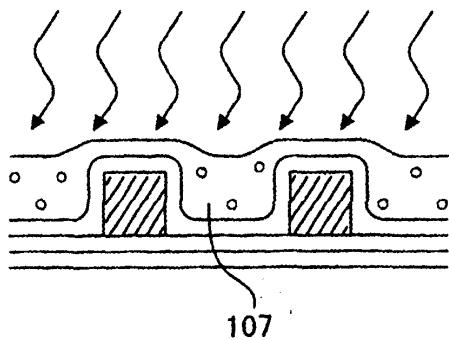


FIG. 1F

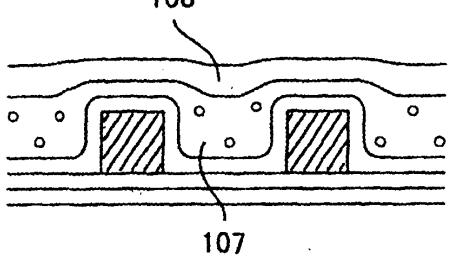


FIG. 1G

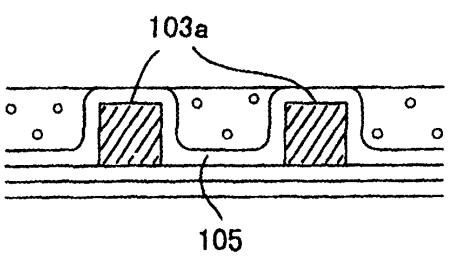


FIG. 1H

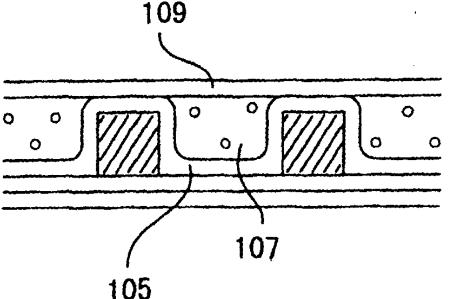


FIG. 2A

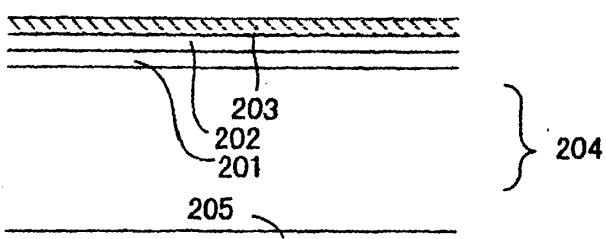


FIG. 2B

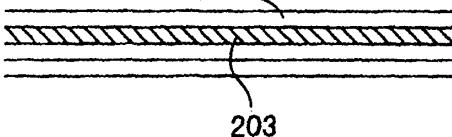


FIG. 2C

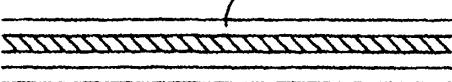
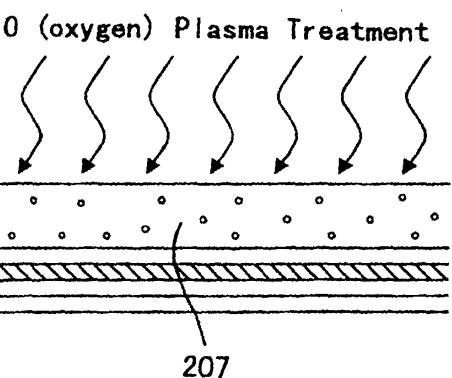


FIG. 2D



H (hydrogen) Plasma Treatment

FIG. 2E

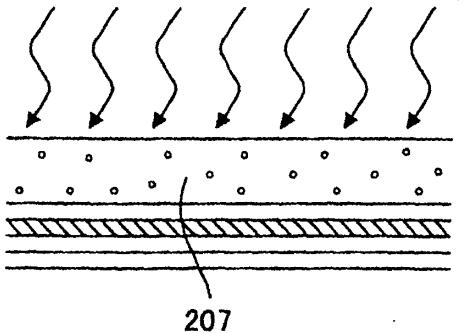


FIG. 2F

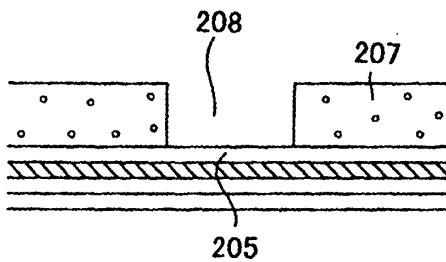


FIG. 2G

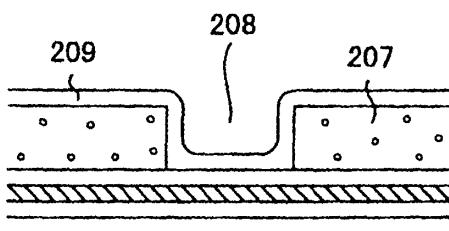


FIG. 2H

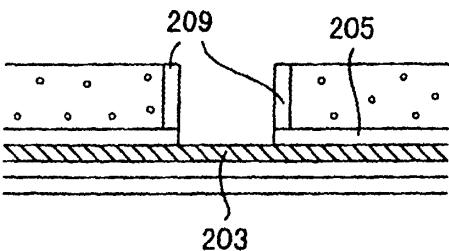


FIG. 2I

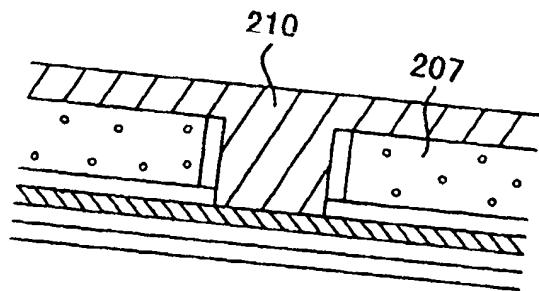


FIG. 2J

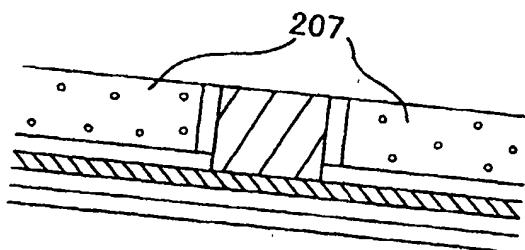


FIG. 2K

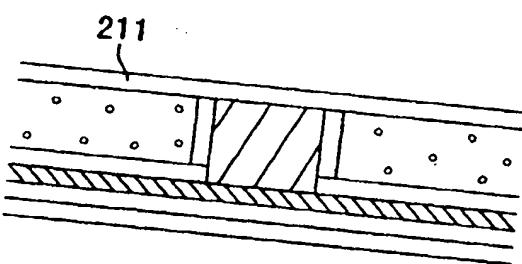


FIG. 2L

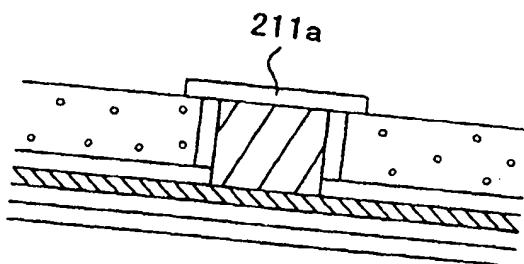
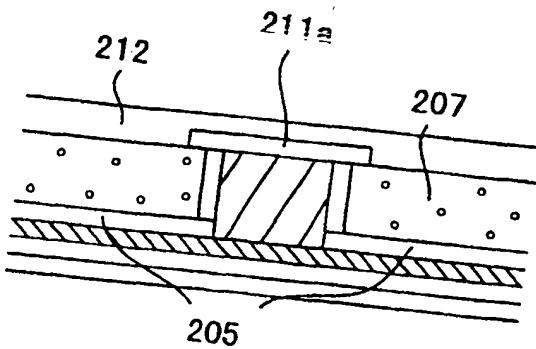


FIG. 2M





| DOCUMENTS CONSIDERED TO BE RELEVANT | | | CLASSIFICATION OF THE APPLICATION (Int.Cl.7) |
|----------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------|---------------------------------------------------------------------------------------------------------------|------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------|-------------------------------------------------|
| Category | Citation of document with indication, where appropriate, of relevant passages | Relevant to claim | |
| X | US 5 494 859 A (KAPOOR ASHOK K) 27 February 1996 (1996-02-27) | 1,4,10, 11,14, 16,19,20 | H01L21/768 H01L21/762 |
| Y | * the whole document * | 2,3,5-9, 12,13, 15,17,18 | |
| X | US 5 744 399 A (PASCH NICHOLAS F ET AL) 28 April 1998 (1998-04-28) * abstract * | 1 | |
| Y | WO 99 12196 A (APPLIED MATERIALS INC) 11 March 1999 (1999-03-11) * the whole document * | 2,5,7,8, 12,13 | |
| Y | EP 0 849 796 A (TEXAS INSTRUMENTS INC) 24 June 1998 (1998-06-24) * column 2, line 6-19; figures 1-9 * | 3,6,9, 17,18 | |
| Y | US 5 614 270 A (LIN SHYUE S ET AL) 25 March 1997 (1997-03-25) * abstract * | 15 | |
| A | GB 2 313 954 A (NIPPON ELECTRIC CO) 10 December 1997 (1997-12-10) * page 17, line 3 - page 18, line 5 * | | TECHNICAL FIELDS SEARCHED (Int.Cl.7) H01L |
| A | US 5 661 344 A (HAVEMANN ROBERT H ET AL) 26 August 1997 (1997-08-26) ----- | | |
| The present search report has been drawn up for all claims | | | |
| Place of search | Date of completion of the search | Examiner | |
| MUNICH | 17 May 2000 | Werner, A | |
| CATEGORY OF CITED DOCUMENTS | | T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons A : member of the same patent family, corresponding document | |
| X : particularly relevant if taken alone Y : particularly relevant if combined with another document of the same category A : technological background O : non-written disclosure P : intermediate document | | | |

EP 1 037 275 A1
ANNEX TO THE EUROPEAN SEARCH REPORT
ON EUROPEAN PATENT APPLICATION NO.

EP 00 10 3333

This annex lists the patent family members relating to the patent documents cited in the above-mentioned European search report. The members are as contained in the European Patent Office EDP file on The European Patent Office is in no way liable for these particulars which are merely given for the purpose of information.

17-05-2000

| Patent document cited in search report | | Publication date | Patent family member(s) | Publication date |
|----------------------------------------|---|------------------|----------------------------------------------|----------------------------------------|
| US 5494859 | A | 27-02-1996 | NONE | |
| US 5744399 | A | 28-04-1998 | NONE | |
| WO 9912196 | A | 11-03-1999 | NONE | |
| EP 0849796 | A | 24-06-1998 | JP 10178006 A | 30-06-1998 |
| US 5614270 | A | 25-03-1997 | NONE | |
| GB 2313954 | A | 10-12-1997 | JP 10056009 A | 24-02-1998 |
| US 5661344 | A | 26-08-1997 | US 5472913 A EP 0703610 A JP 8064680 A | 05-12-1995 27-03-1996 08-03-1996 |

EP001037275A1

For more details about this annex : see Official Journal of the European Patent Office, No. 12/82